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A continuum constraint-free phase field model is proposed to simulate the magnetic domain evolution in ferromagnetic materials. The model takes the polar and azimuthal angles $(\vartheta_1, \vartheta_2)$, instead of the magnetization unit vector $\mathbf{m}(m_1, m_2, m_3)$, as the order parameters. In this way, the constraint on the magnetization magnitude can be exactly satisfied automatically, and no special numerical treatment on the phase field evolution is needed. The phase field model is developed from a thermodynamic framework which involves a configurational force system for ϑ_1 and ϑ_2 . A combination of the configurational force balance and the second law of thermodynamics leads to thermodynamically consistent constitutive relations and a generalized evolution equation for the order parameters $(\vartheta_1, \vartheta_2)$. Beneficial from the constraint-free model, the three-dimensional finite-element implementation is straightforward, and the degrees of freedom are reduced by one. The model is shown to be capable of reproducing the damping-dependent switching dynamics, and the formation and evolution of domains and vortices in ferromagnetic materials under the external magnetic or mechanical loading. Particularly, the calculated out-of-plane component of magnetization in a vortex is verified by the corresponding experimental results, as well as the motion of the vortex under a magnetic field.

1. Introduction

Owing to their ferromagnetic property and magnetic-mechanical coupling, ferromagnetic materials are widely

used in industrial applications, such as magnetic data storage, sensors and actuators, transducers and microelectromechanical systems [1–3]. The viable applications and reasonable design of the devices based on ferromagnetic materials are highly dependent on the fundamental understanding of the microstructures of these materials. At the microscale, ferromagnetic materials are composed of discretized magnetized regions in which the magnetization is uniform. These regions are generally called magnetic domains [4,5]. Accordingly, at the macroscale, the macroscopic magnetic-mechanical properties are determined by these magnetic domain evolutions which can be driven by the external magnetic field and/or mechanical loading [4–6]. Hence, developing a model for ferromagnetic materials that can describe or predict the structure and evolution of the magnetic domains is critically important for understanding, designing and engineering the macroscopic properties of the ferromagnetic devices.

Presently, the micromagnetic model and the phase field method are two widely used approaches for modelling ferromagnetic materials. Based on the well known domain wall calculation by Landau and Lifshitz in 1935 [7], in the 1960s Brown laid the foundation of the micromagnetics theory [8]. The micromagnetic model uses the celebrated Landau–Lifshitz–Gilbert (LLG) equation [9] to describe the temporal evolution of the magnetization and of the domain structure. In the last two decades, the micromagnetic model has achieved vital success in modelling ferromagnetic materials [10–13]. However, in this model, it is difficult to consider the mechanical effects, especially the inhomogeneous stress resulting from the elastic incompatibility of the magnetostrictive strain [14,15]. By contrast, the conventional phase field model is based on the continuum thermodynamics and the kinematics of the materials. This type of model usually uses the magnetization as the order parameters [14–19]. It can easily take into account the magnetic-mechanical coupling and predict the detailed domain structure and evolution under external magnetic field or mechanical loading without *a priori* assumption on domain morphologies.

As in most of the work about micromagnetic models and phase field models for ferromagnetic materials, we consider here merely the evolution at a constant temperature which is far below the Curie temperature and the phase transition temperature. Thereby, the magnetization magnitude remains constant and only its direction changes during the domain evolution. This constraint can be given as

$$\mathbf{M} = M\mathbf{m} \quad \text{with } \|\mathbf{m}\| = 1, \quad (1.1)$$

where \mathbf{M} is the magnetization vector introduced as a continuum field variable, M is the saturation magnetization magnitude and \mathbf{m} is the magnetization unit vector. M is constant when the temperature dependence is ignored. This is intrinsically different from the ferroelectric phase field models with polarization as the order parameter, in which no constraint on the polarization magnitude is enforced [20,21]. In the ferromagnetic materials, the magnetocrystalline anisotropy energy term is non-convex, whose function is similar to that of the Landau energy polynomial in ferroelectrics [20,21]. But it works only when the magnitude constraint of $\|\mathbf{M}\| = M$ is applied. In other words, only when the constraint $\|\mathbf{m}\| = 1$ is considered, this non-convex term can form a multi-well landscape which characterizes the easy axes of the magnetization. This constraint makes the determination of the phase field evolution path challenging and introduces complications in the phase field modelling and numerical implementation [22–25].

In conventional ferromagnetic phase field models, the LLG equation is usually taken as the evolution equation. There the magnitude constraint of $\|\mathbf{M}\| = M$ does not explicitly appear in the formulation but is enforced in the numerical implementation [14,17,19]. This results in a rather simplified presentation. But the numerical implementation of the LLG equation with the constraint is particularly challenging. In the literature, there exist some approximation techniques, such as projection [26], renormalization [14,22], special test function [24] and Lagrange multiplier [23]. There are also exact techniques in use of rotation in the Lie group [25,27]. Recently, Landis introduced a constraint energy term in the form of $A_s(\|\mathbf{M}\| - M)^2$, in order to handle the constraint in an energetic formulation [18]. However, in this way, the constraint is only fulfilled at the stationary solution, and the evolution dynamics is not physically sound. Furthermore, it is not

so easy to choose a reasonable value for the constraint energy coefficient A_s . Wang & Zhang [15] have used this constraint energy proposed by Landis and the time-dependent Ginzburg–Landau (TDGL) equation which is widely used in ferroelectrics to establish their phase field model. It should be noted that if the LLG equation instead of the TDGL equation is used, the contribution of the constraint energy to the LLG equation vanishes; because the equality $\mathbf{m} \times \mathbf{m} = \mathbf{0}$ always holds in the LLG equation. Therefore, the method of constraint energy is not so effective.

In this work, we develop a novel constraint-free phase field model for mechanically coupled magnetic domain evolution in ferromagnetic materials. The focus here is to construct a model which delivers the correct evolution dynamics and fulfils the constraint automatically. The idea is to take the polar and azimuthal angles $(\vartheta_1, \vartheta_2)$, instead of the magnetization unit vector $\mathbf{m}(m_1, m_2, m_3)$, as the order parameters. It is apparent that the Cartesian components of \mathbf{m} can be expressed by $(\vartheta_1, \vartheta_2)$ in a constraint-free formulation, i.e. $m_1 = \sin \vartheta_1 \cos \vartheta_2$, $m_2 = \sin \vartheta_1 \sin \vartheta_2$ and $m_3 = \cos \vartheta_1$. By using the configurational force theory and the second law of thermodynamics, a set of thermodynamically consistent constitutive relations and a generalized evolution equation for ϑ_1 and ϑ_2 are derived, as shown in §2. In §3, a three-dimensional nonlinear finite-element formulation of this constraint-free model is presented. Thereby no additional numerical technique is required for the magnitude constraint of $\|\mathbf{m}\| = 1$. Furthermore, the node degrees of freedom are decreased by one. In §4, several numerical examples are presented. Benchmark test on precession and precessional switching is checked in the use of one-element simulation, and it demonstrates that the model reproduces the correct magnetization dynamics. Modelling on domain formation and evolution, such as vortices and 180° domain switching, shows that three-dimensional simulations are required in order to recapitulate the spatial switching path. The shape and motion of the computed vortices agree well with the related experimental results. In addition, the ferroelastic switching is also simulated. In comparison with other phase field simulations, the proposed three-dimensional constraint-free model has the merit that it can readily reveal the damping-dependent and the out-of-plane of magnetization in a thin film.

2. Continuum constraint-free phase field model

(a) Field equations

The magnetoelastic coupling is one of the most important properties for ferromagnetic materials. Thus both the mechanics and the magnetostatics should be considered in the modelling. For a ferromagnetic body \mathcal{B} with a boundary $\partial\mathcal{B}$, the quasi-static mechanical equilibrium is described by

$$\sigma_{ij,j} + f_i = 0 \quad \text{in } \mathcal{B}, \quad (2.1)$$

where σ_{ij} is the Cauchy stress and f_i is the body force. Hereafter, the Latin indices (i, j, k, l, p, q) run over the range of 1–3. The comma in a subscript denotes spatial differentiation, for example, $\sigma_{ij,j} = \partial\sigma_{ij}/\partial x_j$ in which x_j is the j th Cartesian coordinate direction. The Einstein summation convention is used for the repeated indices. Two types of boundary conditions can be introduced

$$u_i = \hat{u}_i \quad \text{on } \partial\mathcal{B}_u \quad \text{and} \quad \sigma_{ij}n_j = \hat{t}_i \quad \text{on } \partial\mathcal{B}_\sigma \quad (2.2)$$

in which \hat{u}_i is the displacement prescribed on the boundary part $\partial\mathcal{B}_u$, n_j is the outward surface unit vector and \hat{t}_i is the surface traction on the boundary part $\partial\mathcal{B}_\sigma$. By assuming linear kinematics, the strain ε_{ij} can be expressed as the symmetrical gradient of the displacement field u_i

$$\varepsilon_{ij} = \frac{1}{2}(u_{i,j} + u_{j,i}). \quad (2.3)$$

The Maxwell equation which governs the magnetic part has the form

$$B_{i,i} = 0 \quad \text{in } \mathcal{B}, \quad (2.4)$$

where B_i is the magnetic induction. As the permittivity of the ferromagnetic materials is usually much higher than that of the free space, the stray field effect is ignored here. Thus the magnetic

boundary conditions can take the form of

$$B_i n_i = \hat{B} \quad \text{on } \partial \mathcal{B}_B \quad \text{and} \quad \phi = \hat{\phi} \quad \text{on } \partial \mathcal{B}_\phi, \quad (2.5)$$

where \hat{B} is the prescribed value on the boundary part $\partial \mathcal{B}_B$, ϕ is the scalar magnetic potential and $\hat{\phi}$ is the given potential on the boundary part $\partial \mathcal{B}_\phi$. The magnetic field H_i is defined by the negative gradient of ϕ

$$H_i = -\phi_{,i}. \quad (2.6)$$

(b) Balance law for magnetization

The magnetic-mechanical couple in ferromagnetic materials results from the existence and rearrangement of ferromagnetic domains [4]. The free energy of the ferromagnetic materials is also dependent on the magnetization configuration. In the continuum theory, the magnetization configuration can be described by the distribution of the magnetization vector. To obtain the distribution, one needs to first consider the balance law of magnetization. Though the magnetization dynamics has been well established [8], in this subsection we employ the configurational force theory [28–30] to represent the balance law, in order to support the derivation of our phase field model in the following subsections.

As it has been mentioned, we consider only the isothermal process below the Currie temperature and the phase transition temperature. Thus the magnitude of the magnetization vector remains constant, and the configuration can be depicted by the unit magnetization vector m_i . The configurational force system for the configurational quantity m_i includes the configurational stress tensor Σ_{ij} whose power density expended on the surface is $\Sigma_{ij} n_j M \dot{m}_i$, and the internal and external configurational force vector g_i and g_i^{ex} whose power density expended in the volume is $g_i M \dot{m}_i$ and $g_i^{\text{ex}} M \dot{m}_i$, respectively. Owing to the rotation nature of the magnetization [9], the angular momentum balance in the continuum aspect can be given as

$$\int_{\mathcal{B}} \dot{m}_i \, dv = \gamma_0 \left(\int_{\partial \mathcal{B}} \epsilon_{ijk} \Sigma_{jl} n_l m_k \, ds + \int_{\mathcal{B}} \epsilon_{ijk} g_j m_k \, dv + \int_{\mathcal{B}} \epsilon_{ijk} g_j^{\text{ex}} m_k \, dv \right), \quad (2.7)$$

where γ_0 is the gyromagnetic ratio with a positive constant value of $1.76 \times 10^{11} / (Ts)$, and ϵ_{ijk} is the permutation tensor. Converting the surface integration in (2.7) into volume integration and considering its validity in any volume, we can obtain

$$\frac{1}{\gamma_0} \dot{m}_i = \epsilon_{ijk} \Sigma_{jl,l} m_k + \epsilon_{ijk} \Sigma_{jl} m_{k,l} + \epsilon_{ijk} g_j m_k + \epsilon_{ijk} g_j^{\text{ex}} m_k. \quad (2.8)$$

The magnetization can only change its direction, so the configurational force, which physically is the driving force on the change of magnetization, should lie perpendicular to the direction of the magnetization. In other words, the configurational force along the magnetization direction must be zero. Thus, in any volume, we have

$$m_i \left(\int_{\partial \mathcal{B}} \Sigma_{ij} n_j \, ds + \int_{\mathcal{B}} g_i \, dv + \int_{\mathcal{B}} g_i^{\text{ex}} \, dv \right) = 0 \quad \Rightarrow \quad m_i (\Sigma_{ij,j} + g_i + g_i^{\text{ex}}) = 0. \quad (2.9)$$

From (2.8) and (2.9), one has

$$\Sigma_{ij,j} + g_i + g_i^{\text{ex}} = \frac{1}{\gamma_0} \epsilon_{ijk} m_j \dot{m}_k + \mathcal{M}_i \quad (2.10)$$

with $\mathcal{M}_i = \epsilon_{ijk} \epsilon_{jpq} \Sigma_{pl} m_{q,l} m_k$.

(c) Thermodynamics

As pointed out in the Introduction, although the use of $\mathbf{m}(m_1, m_2, m_3)$ as the order parameters is straightforward, the numerical implementation of the constraint on the magnetization magnitude can be complicated [22–25]. Herein, we use the polar and azimuthal angles $(\vartheta_1, \vartheta_2)$ as order

parameters. Thus the components of the unit magnetization vector in the corresponding Cartesian coordinates are given as

$$\mathbf{m} = \mathbf{m}(\vartheta_1, \vartheta_2) = \begin{bmatrix} \sin \vartheta_1 \cos \vartheta_2 \\ \sin \vartheta_1 \sin \vartheta_2 \\ \cos \vartheta_1 \end{bmatrix} \quad (2.11)$$

and

$$\dot{m}_i = A_{\mu i} \dot{\vartheta}_\mu \quad \text{with } A_{\mu i} = \begin{bmatrix} \cos \vartheta_1 \cos \vartheta_2 & -\sin \vartheta_1 \sin \vartheta_2 \\ \cos \vartheta_1 \sin \vartheta_2 & \sin \vartheta_1 \cos \vartheta_2 \\ -\sin \vartheta_1 & 0 \end{bmatrix}. \quad (2.12)$$

In this paper, the Greek indices μ and γ run over the range of 1–2. The constraint $\|\mathbf{m}\| = 1$ is satisfied automatically. The number of order parameters is also reduced from 3 to 2.

For the temperature-independent process, the Helmholtz free energy of the ferromagnetic system with magnetic-mechanical couple can be taken as $\mathcal{F} = \tilde{\mathcal{F}}(\varepsilon_{ij}, B_i, \vartheta_\mu, \vartheta_{\mu,i})$. According to the second law of thermodynamics, under the isothermal condition, the external power expended on the control volume should not be less than the change rate in the Helmholtz free energy, i.e.

$$\int_{\partial \mathcal{B}} [\sigma_{ij} n_j \dot{u}_i - \phi \dot{B}_i n_i + \Sigma_{ij} n_j M A_{\mu i} \dot{\vartheta}_\mu] ds + \int_{\mathcal{B}} [f_i \dot{u}_i + g_i^{\text{ex}} M A_{\mu i} \dot{\vartheta}_\mu] dv \geq \int_{\mathcal{B}} \dot{\mathcal{F}} dv. \quad (2.13)$$

In this inequality, the internal configurational force g_i is omitted as it has no external power. The difference between the right-hand and left-hand terms is the dissipation. After the Legendre transformation, the magnetic enthalpy $\mathcal{H} = \tilde{\mathcal{H}}(\varepsilon_{ij}, H_i, \vartheta_\mu, \vartheta_{\mu,i})$ can be derived as $\mathcal{H} = \mathcal{F} - B_i H_i$. Thus, the thermodynamic inequality (2.13) can be rewritten as

$$\begin{aligned} & \int_{\partial \mathcal{B}} [\sigma_{ij} n_j \dot{u}_i - \phi \dot{B}_i n_i + \Sigma_{ij} n_j M A_{\mu i} \dot{\vartheta}_\mu] ds + \int_{\mathcal{B}} [f_i \dot{u}_i + g_i^{\text{ex}} M A_{\mu i} \dot{\vartheta}_\mu] dv \\ & \geq \int_{\mathcal{B}} (\dot{\mathcal{H}} + B_i \dot{H}_i + H_i \dot{B}_i) dv \end{aligned} \quad (2.14)$$

or by the Gauss law

$$\int_{\mathcal{B}} [(\sigma_{ij,j} + f_i) \dot{u}_i + \sigma_{ij} \dot{\varepsilon}_{ij} - \phi \dot{B}_{i,i} + M \Pi_{\mu j} \dot{\vartheta}_{\mu,j} + M (\Pi_{\mu j,j} + \zeta_\mu^{\text{ex}}) \dot{\vartheta}_\mu] dv \geq \int_{\mathcal{B}} (\dot{\mathcal{H}} + B_i \dot{H}_i) dv \quad (2.15)$$

with

$$\Pi_{\mu j} = A_{\mu i} \Sigma_{ij} \quad \text{and} \quad \zeta_\mu^{\text{ex}} = A_{\mu i} g_i^{\text{ex}} \quad (2.16)$$

Here, $\Pi_{\mu j}$ and ζ_μ^{ex} are the configurational stress tensor and the external configurational force, with respect to the order parameters ϑ_μ , respectively.

Left multiplication of (2.10) by matrix $A_{\mu i}$ leads to

$$A_{\mu i} \Sigma_{ij,j} + \zeta_\mu + \zeta_\mu^{\text{ex}} = \frac{1}{\gamma_0} \sin \vartheta_1 \mathcal{T}_{\mu\gamma} \dot{\vartheta}_\gamma + A_{\mu i} \mathcal{M}_i \quad (2.17)$$

in which $\zeta_\mu = A_{\mu i} g_i$ is the internal configurational force with respect to ϑ_μ , and $\mathcal{T}_{\mu\gamma}$ is the antisymmetric matrix

$$\mathcal{T}_{\mu\gamma} = \begin{bmatrix} 0 & -1 \\ 1 & 0 \end{bmatrix}. \quad (2.18)$$

Application of (2.17) and the form of \mathcal{H} to (2.15) leads to

$$\begin{aligned} & \int_{\mathcal{B}} \left\{ (\sigma_{ij,j} + f_i) \dot{u}_i - \phi \dot{B}_{i,i} + \left(\sigma_{ij} - \frac{\partial \mathcal{H}}{\partial \varepsilon_{ij}} \right) \dot{\varepsilon}_{ij} + M \left[A_{\mu i} \mathcal{M}_i - A_{\mu i} \Sigma_{ij,j} - \zeta_\mu + \Pi_{\mu j,j} - \frac{1}{M} \frac{\partial \mathcal{H}}{\partial \vartheta_\mu} \right] \dot{\vartheta}_\mu \right. \\ & \left. - \left(B_i + \frac{\partial \mathcal{H}}{\partial H_i} \right) \dot{H}_i + M \left(\Pi_{\mu j} - \frac{1}{M} \frac{\partial \mathcal{H}}{\partial \vartheta_{\mu,j}} \right) \dot{\vartheta}_{\mu,j} \right\} dv \geq 0. \end{aligned} \quad (2.19)$$

Considering the field equations (2.1) and (2.4) and noticing that inequality (2.19) must be hold for all admissible processes, based on the standard arguments of rational thermomechanics, we can

obtain these constitutive relations

$$\sigma_{ij} = \frac{\partial \mathcal{H}}{\partial \varepsilon_{ij}}, \quad B_i = -\frac{\partial \mathcal{H}}{\partial H_i} \quad \text{and} \quad \Pi_{\mu j} = \frac{1}{M} \frac{\partial \mathcal{H}}{\partial \vartheta_{\mu j}}. \quad (2.20)$$

Given these relations, the residual dissipation inequality in (2.19) in the local form can be derived as

$$-g_{\mu}^{\text{dis}} \dot{\vartheta}_{\mu} \geq 0, \quad (2.21)$$

where

$$g_{\mu}^{\text{dis}} = -A_{\mu i} \mathcal{M}_i + A_{\mu i} \Sigma_{ijj} + \zeta_{\mu} - \Pi_{\mu jj} + \pi_{\mu}, \quad \pi_{\mu} = \frac{1}{M} \frac{\partial \mathcal{H}}{\partial \vartheta_{\mu}} \quad (2.22)$$

The residual dissipation inequality (2.21) is satisfied by setting

$$g_{\mu}^{\text{dis}} = -\beta_{\mu\gamma} \dot{\vartheta}_{\gamma}, \quad (2.23)$$

where $\beta_{\mu\gamma}$ is the components of a positive semi-definite matrix which can be derived through a non-concave dissipation potential $\mathcal{D}(\vartheta_{\mu})$

$$\beta_{\mu\gamma} = \frac{\partial^2 \mathcal{D}}{\partial \dot{\vartheta}_{\mu} \partial \dot{\vartheta}_{\gamma}}. \quad (2.24)$$

Combining (2.17), (2.22) and (2.23), we can obtain a generalized form of the evolution equation for the order parameters ϑ_{μ} as

$$\Pi_{\mu jj} - \pi_{\mu} + \zeta_{\mu}^{\text{ex}} = \left(\frac{1}{\gamma_0} \sin \vartheta_1 \mathcal{I}_{\mu\gamma} + \beta_{\mu\gamma} \right) \dot{\vartheta}_{\gamma}. \quad (2.25)$$

It should be noted that when the Rayleigh dissipation functional is adopted, (2.25) can be reduced to the standard LLG equation. In fact, inserting the Rayleigh dissipation potential [9]

$$\mathcal{D} = \frac{\nu}{2} \dot{m}_i \dot{m}_i = \frac{\nu}{2} (\dot{\vartheta}_1^2 + \sin^2 \vartheta_1 \dot{\vartheta}_2^2) \quad (2.26)$$

into (2.24), one has

$$\beta_{\mu\gamma} = \nu P_{\mu\gamma} \quad \text{with} \quad P_{\mu\gamma} = \begin{bmatrix} 1 & 0 \\ 0 & \sin^2 \vartheta_1 \end{bmatrix}. \quad (2.27)$$

Insertion of (2.27) into (2.25) leads to

$$\Pi_{\mu jj} - \pi_{\mu} + \zeta_{\mu}^{\text{ex}} = \frac{1}{\gamma_0} L_{\mu\gamma} \dot{\vartheta}_{\gamma}, \quad (2.28)$$

where

$$L_{\mu\gamma} = \begin{bmatrix} \alpha & -\sin \vartheta_1 \\ \sin \vartheta_1 & \alpha \sin^2 \vartheta_1 \end{bmatrix} \quad (2.29)$$

and $\alpha = \gamma_0 \nu$ is the damping coefficient which generally appears in the LLG equation. As it is well known, the LLG equation is given in terms of \mathbf{m} in the following form:

$$\dot{\mathbf{m}} = -\gamma_0 \mu_0 \mathbf{m} \times \mathbf{H}^{\text{eff}} + \alpha \mathbf{m} \times \dot{\mathbf{m}}, \quad (2.30)$$

where \mathbf{H}^{eff} is the effective field obtained by variational derivation of the total magnetic enthalpy with respect to \mathbf{m} , μ_0 is the vacuum permeability constant and \times denotes the cross product of two vectors. By using the relation $\mathbf{m} = (\sin \vartheta_1 \cos \vartheta_2, \sin \vartheta_1 \sin \vartheta_2, \cos \vartheta_1)$, it is not difficult to prove that (2.30) is equivalent to (2.28). For more details, one is referred to the appendix A.

(d) Magnetic enthalpy

In the micromagnetic model, the free energy determined by the magnetization configuration contains the elastic contribution \mathcal{H}^{ela} , the magnetocrystalline anisotropy contribution \mathcal{H}^{ani} , the

exchange contribution \mathcal{H}^{exc} and the magnetostatic contribution \mathcal{H}^{mag} . For a cubic crystal, the following magnetic enthalpy is used

$$\mathcal{H} = \mathcal{H}^{\text{ela}} + \mathcal{H}^{\text{ani}} + \mathcal{H}^{\text{exc}} + \mathcal{H}^{\text{mag}} \quad (2.31)$$

with

$$\left. \begin{aligned} \mathcal{H}^{\text{ela}} &= \frac{1}{2} C_{ijkl} (\varepsilon_{ij} - \varepsilon_{ij}^0) (\varepsilon_{kl} - \varepsilon_{kl}^0), \\ \mathcal{H}^{\text{ani}} &= K_1 (m_1^2 m_2^2 + m_2^2 m_3^2 + m_3^2 m_1^2) + K_2 m_1^2 m_2^2 m_3^2 \\ &= K_1 (\sin^4 \vartheta_1 \sin^2 \vartheta_2 \cos^2 \vartheta_2 + \sin^2 \vartheta_1 \cos^2 \vartheta_1) \\ &\quad + K_2 \sin^4 \vartheta_1 \cos^2 \vartheta_1 \sin^2 \vartheta_2 \cos^2 \vartheta_2, \\ \mathcal{H}^{\text{exc}} &= A_e m_{i,j} m_{i,j} = A_e (\vartheta_{1,j} \vartheta_{1,j} + \sin^2 \vartheta_1 \vartheta_{2,j} \vartheta_{2,j}) \\ \mathcal{H}^{\text{mag}} &= -\frac{1}{2} \mu_0 H_i H_i - \mu_0 M H_i m_i(\vartheta_\mu), \end{aligned} \right\} \quad (2.32)$$

and

where C_{ijkl} is the component of the material elastic tensor, A_e is the exchange constant, and K_1 and K_2 are the magnetocrystalline anisotropy constants. Spontaneous magnetization-induced strain ε_{ij}^0 for cubic crystals, can be described as

$$\left. \begin{aligned} \varepsilon_{11}^0 &= \frac{3}{2} \lambda_{100} (\sin^2 \vartheta_1 \cos^2 \vartheta_2 - \frac{1}{3}), & \varepsilon_{12}^0 = \varepsilon_{21}^0 &= \frac{3}{4} \lambda_{111} \sin^2 \vartheta_1 \sin 2\vartheta_2 \\ \varepsilon_{22}^0 &= \frac{3}{2} \lambda_{100} (\sin^2 \vartheta_1 \sin^2 \vartheta_2 - \frac{1}{3}), & \varepsilon_{23}^0 = \varepsilon_{32}^0 &= \frac{3}{4} \lambda_{111} \sin 2\vartheta_1 \sin \vartheta_2 \\ \varepsilon_{33}^0 &= \frac{3}{2} \lambda_{100} (\cos^2 \vartheta_1 - \frac{1}{3}), & \varepsilon_{31}^0 = \varepsilon_{13}^0 &= \frac{3}{4} \lambda_{111} \sin 2\vartheta_1 \cos \vartheta_2, \end{aligned} \right\} \quad (2.33)$$

and

where λ_{100} and λ_{111} is the magnetostriction strain along the direction $\langle 100 \rangle$ and $\langle 111 \rangle$ of a single crystal, respectively, when it is magnetized at saturation along this direction.

With the above-specified magnetic enthalpy, the constitutive relations in (2.20) can be given as

$$\left. \begin{aligned} \sigma_{ij} &= C_{ijkl} (\varepsilon_{kl} - \varepsilon_{kl}^0), \\ B_i &= \mu_0 [H_i + M m_i(\vartheta_\mu)] \\ \Pi_{\mu j} &= \frac{2A_e}{M} P_{\mu\gamma} \vartheta_{\gamma j}. \end{aligned} \right\} \quad (2.34)$$

and

The evolution equation (2.28) can be derived as

$$\Pi_{\mu j,j} - \frac{1}{M} \frac{\partial \mathcal{H}}{\partial \vartheta_\mu} + \zeta_\mu^{\text{ex}} = \frac{1}{\gamma_0} L_{\mu\gamma} \dot{\vartheta}_\gamma, \quad (2.35)$$

where

$$\frac{\partial \mathcal{H}}{\partial \vartheta_\mu} = -\frac{\partial \varepsilon_{ij}^0}{\partial \vartheta_\mu} \sigma_{ij} + \frac{\partial \mathcal{H}^{\text{ani}}}{\partial \vartheta_\mu} + A_e Q_\mu - \mu_0 M A_{\mu i} H_i, \quad \text{with } Q_\mu = \begin{bmatrix} \sin 2\vartheta_1 \vartheta_{2,j} \vartheta_{2,j} \\ 0 \end{bmatrix}. \quad (2.36)$$

The boundary condition and initial condition for ϑ can be set as

$$\vartheta_\mu = \hat{\vartheta}_\mu \quad \text{on } \partial \mathcal{B}_\vartheta, \quad \vartheta_{\mu,j} n_j = \hat{\vartheta}_\mu \quad \text{on } \partial \mathcal{B}_\varnothing \quad \text{and} \quad \vartheta_\mu(x_i, t)|_{t=0} = \hat{\vartheta}_\mu^0(x_i) \quad \text{in } \mathcal{B}. \quad (2.37)$$

The combination of (2.31)–(2.36) constitutes the constraint-free formulation of the continuum phase field model for ferromagnetic materials. No additional constraint on the magnitude of \mathbf{M} is required, because $\|\mathbf{M}\| = M$ has been intrinsically implemented in the model.

3. Finite-element implementation

(a) Dimensionless form

A direct implementation of the constrain-free model in the physical dimension leads to equation systems with large condition numbers in the order of 10^{17} . This can be due to the varying orders of the parameters in the magnetic and the mechanical problem. For example, the elastic constant is in

the order of 10^{11} , whereas the vacuum permeability is in the order of 10^{-6} . Hence, a dimensionless form of the constrain-free model is implemented here. The model is normalized via the following ansatz

$$\left. \begin{aligned} K_1^* &= 1, \quad \mu_0^* = 1, \quad A_e^* = 1, \quad \mathcal{H}^* = \frac{\mathcal{H}}{K_1}, \quad K_2^* = \frac{K_2}{K_1}, \quad t^* = \frac{\gamma_0 K_1 t}{M} \\ x_i^* &= x_i \sqrt{\frac{K_1}{A_e}}, \quad ()_{,i^*} = \frac{\partial ()}{\partial x_i^*}, \quad u_i^* = u_i \sqrt{\frac{K_1}{A_e}}, \quad M^* = M \sqrt{\frac{\mu_0}{K_1}} \\ C_{ijkl}^* &= \frac{C_{ijkl}}{K_1}, \quad \sigma_{ij}^* = \frac{\sigma_{ij}}{K_1}, \quad \varepsilon_{ij}^* = \varepsilon_{ij}, \quad \varepsilon_{ij}^{0*} = \varepsilon_{ij}^0, \quad H_i^* = H_i \sqrt{\frac{\mu_0}{K_1}} \\ B_i^* &= \frac{B_i}{\sqrt{K_1 \mu_0}}, \quad \dot{\vartheta}_\mu^* = \frac{\dot{\vartheta}_\mu M}{(K_1 \gamma_0)}, \quad \phi^* = \phi \sqrt{\frac{\mu_0}{A_e}}, \quad f_i^* = \frac{f_i \sqrt{A_e/K_1}}{K_1} \\ \zeta_\mu^{\text{ex}*} &= \frac{\zeta_\mu^{\text{ex}} M}{K_1}. \end{aligned} \right\} \quad (3.1)$$

and

The magnetic enthalpy, the field equations, the constitute relations and the evolution equation take the corresponding dimensionless form as follows:

$$\begin{aligned} \mathcal{H}^* &= \frac{1}{2} C_{ijkl}^* (\varepsilon_{ij}^* - \varepsilon_{ij}^{0*}) (\varepsilon_{kl}^* - \varepsilon_{kl}^{0*}) + m_1^2 m_2^2 + m_2^2 m_3^2 + m_3^2 m_1^2 + K_2^* m_1^2 m_2^2 m_3^2 \\ &\quad + \vartheta_{1,j^*} \vartheta_{1,j^*} + \sin^2 \vartheta_1 \vartheta_{2,j^*} \vartheta_{2,j^*} - \frac{1}{2} H_i^* H_i^* - M^* H_i^* m_i \end{aligned} \quad (3.2)$$

$$\sigma_{ij,j^*}^* + f_i^* = 0 \quad \text{in } \mathcal{B} \quad (3.3)$$

$$B_{i,j^*}^* = 0 \quad \text{in } \mathcal{B} \quad (3.4)$$

$$\varepsilon_{ij}^* = \frac{1}{2} (u_{i,j^*}^* + u_{j,i^*}^*), \quad H_i^* = -\phi_{,i^*}^* \quad \text{in } \mathcal{B} \quad (3.5)$$

$$\sigma_{ij}^* = C_{ijkl}^* (\varepsilon_{kl}^* - \varepsilon_{kl}^{0*}), \quad B_i^* = H_i^* + M^* m_i, \quad \Pi_{\mu j}^* = 2P_{\mu\gamma} \vartheta_{\gamma j^*} \quad (3.6)$$

$$\text{and} \quad \Pi_{\mu j,j^*}^* - \frac{\partial \mathcal{H}^*}{\partial \vartheta_\mu} + \zeta_\mu^{\text{ex}*} = L_{\mu\gamma} \dot{\vartheta}_\gamma^* \quad (3.7)$$

The dimensionless boundary and initial conditions are

$$\left. \begin{aligned} u_i^* &= \hat{u}_i^* \quad \text{on } \partial \mathcal{B}_u, \quad \sigma_{ij}^* n_j^* = \hat{t}_i^* \quad \text{on } \partial \mathcal{B}_\sigma, \\ B_i^* n_i^* &= \hat{B}^* \quad \text{on } \partial \mathcal{B}_B, \quad \phi^* = \hat{\phi}^* \quad \text{on } \partial \mathcal{B}_\phi, \\ \vartheta &= \hat{\vartheta} \quad \text{on } \partial \mathcal{B}_\vartheta, \quad \vartheta_{\mu,j^*} n_j^* = \hat{\Theta}_\mu^* \quad \text{on } \partial \mathcal{B}_\Theta \end{aligned} \right\} \quad (3.8)$$

and

$$\vartheta_\mu(x_i^*, t^*)|_{t^*=0} = \hat{\vartheta}_\mu^0(x_i^*) \quad \text{in } \mathcal{B}$$

Here, $\hat{\Theta}_\mu^*$ is the boundary value defined for ϑ_{μ,j^*} , as it is shown in (2.37).

Remark. In the next section for the finite element formulation, only the dimensionless form of the phase field model will be employed. The superscript stars are hence omitted for notational simplicity.

(b) Finite-element formulation

The presented phase field model in the constraint-free form is implemented in the three-dimensional finite-element method. The discretization is achieved by eight-node linear elements. For more details, readers are referred to standard textbooks on finite-element methods [31]. The displacement $\mathbf{u}(u_1, u_2, u_3)$, the scalar magnetic potential ϕ , and the polar and azimuthal angles $\vartheta(\vartheta_1, \vartheta_2)$ are taken as independent variables. Thus, each node has six degrees of freedom $\underline{\mathbf{d}}^I = [u_1^I \ u_2^I \ u_3^I \ \phi^I \ \vartheta_1^I \ \vartheta_2^I]^T$, where the superscript I indicates the element node and the underbar denotes a matrix. Note that if \mathbf{m} is taken as the order parameter, it requires not only additional constrain of $\|\mathbf{m}\| = 1$, but also one more degree of freedom. In this sense, the constraint-free formulation favours an efficient numerical implementation.

The body force f_i in (3.3) and the external configurational force ζ_μ^{ex} in (3.7) are neglected. Actually, ζ_μ^{ex} can be given by setting the boundary condition of ϕ . With these simplicities, corresponding to the strong forms in (3.3), (3.4) and (3.7), the following three weak forms could be formulated

$$\left. \begin{aligned} 0 &= - \int_{\mathcal{B}} \sigma_{ij} \eta_{i,j}^u \, dv + \int_{\partial \mathcal{B}} \hat{t}_i \eta_i^u \, ds, \\ 0 &= - \int_{\mathcal{B}} B_i \eta_i^\phi \, dv + \int_{\partial \mathcal{B}} \hat{B} \eta^\phi \, ds \\ \text{and} \quad 0 &= - \int_{\mathcal{B}} \left[\left(L_{\mu\gamma} \vartheta_\gamma + \frac{\partial \mathcal{H}}{\partial \vartheta_\mu} \right) \eta_\mu^\vartheta + \Pi_{\mu j} \eta_{\mu,j}^\vartheta \right] dv + \int_{\partial \mathcal{B}} 2P_{\mu\gamma} \hat{\Theta}_\gamma \eta_\mu^\vartheta \, ds, \end{aligned} \right\} \quad (3.9)$$

where η_i^u , η^ϕ and η_μ^ϑ are the test functions for u_i , ϕ , and ϑ_μ , respectively.

By introducing the shape functions for the independent variables, the rate of the polar and azimuthal angles, and the test functions, the discretized equations are obtained as

$$\left. \begin{aligned} \underline{\mathbf{u}} &= \sum_I N_u^I \underline{\mathbf{u}}^I, \quad \underline{\boldsymbol{\eta}}^u = \sum_I N_u^I \underline{\boldsymbol{\eta}}^{uI}, \quad \phi = \sum_I N_\phi^I \phi^I, \quad \eta^\phi = \sum_I N_\phi^I \eta^{\phi I} \\ \text{and} \quad \underline{\boldsymbol{\vartheta}} &= \sum_I N_\vartheta^I \underline{\boldsymbol{\vartheta}}^I, \quad \underline{\dot{\boldsymbol{\vartheta}}} = \sum_I N_\vartheta^I \underline{\dot{\boldsymbol{\vartheta}}}^I, \quad \underline{\boldsymbol{\eta}}^\vartheta = \sum_I N_\vartheta^I \underline{\boldsymbol{\eta}}^{\vartheta I}. \end{aligned} \right\} \quad (3.10)$$

The superscript I denotes the node number. N_u^I , N_ϕ^I and N_ϑ^I are the shape functions for the displacements, the scalar magnetic potential and the polar and azimuthal angles, respectively. In this paper, the underbar denotes Voigt notation of the corresponding quantities. Accordingly, the strain in (2.3), the magnetic field in (2.6) and $\eta_{\mu,j}^\vartheta$ can be given as

$$\underline{\boldsymbol{\varepsilon}} = \sum_I \underline{\mathbf{B}}_u^I \underline{\mathbf{u}}^I, \quad \underline{\mathbf{H}} = - \sum_I \underline{\mathbf{B}}_\phi^I \phi^I \quad \text{and} \quad \nabla \underline{\boldsymbol{\eta}}^\vartheta = \sum_I \underline{\mathbf{B}}_\vartheta^I \underline{\boldsymbol{\eta}}^{\vartheta I} \quad (3.11)$$

and the constitutive relations in (3.6) as

$$\left. \begin{aligned} \underline{\boldsymbol{\sigma}} &= \underline{\mathbf{C}}(\underline{\boldsymbol{\varepsilon}} - \underline{\boldsymbol{\varepsilon}}^0), \quad \underline{\mathbf{B}} = \underline{\mathbf{H}} + M \underline{\mathbf{m}} \\ \text{and} \quad \underline{\boldsymbol{\Pi}} &= [\vartheta_{1,1} \quad \sin^2 \vartheta_{1,2} \vartheta_{1,2} \quad \vartheta_{1,2} \quad \sin^2 \vartheta_{1,2,1} \quad \vartheta_{1,3} \quad \sin^2 \vartheta_{1,2,3}]^T. \end{aligned} \right\} \quad (3.12)$$

In the three-dimensional case, the matrices in (3.11) can be given as

$$\underline{\mathbf{B}}_u^I = \begin{bmatrix} N_{u,1}^I & 0 & 0 \\ 0 & N_{u,2}^I & 0 \\ 0 & 0 & N_{u,3}^I \\ N_{u,2}^I & N_{u,1}^I & 0 \\ 0 & N_{u,3}^I & N_{u,2}^I \\ N_{u,3}^I & 0 & N_{u,1}^I \end{bmatrix}, \quad \underline{\mathbf{B}}_\phi^I = \begin{bmatrix} N_{\vartheta,1}^I & 0 \\ 0 & N_{\vartheta,2}^I \\ N_{\vartheta,2}^I & 0 \\ 0 & N_{\vartheta,1}^I \\ N_{\vartheta,3}^I & 0 \\ 0 & N_{\vartheta,3}^I \end{bmatrix} \quad \text{and} \quad \underline{\mathbf{B}}_\vartheta^I = \begin{bmatrix} N_{\phi,1}^I \\ N_{\phi,2}^I \\ N_{\phi,3}^I \end{bmatrix}. \quad (3.13)$$

Inserting the (3.10)–(3.13) into the weak forms (3.9) and taking the integration over the volume \mathcal{B}_e of one element, we can obtain the element residuals

$$\left. \begin{aligned} \underline{\mathbf{R}}_u^I &= - \int_{\mathcal{B}_e} (\underline{\mathbf{B}}_u^I)^T \underline{\boldsymbol{\sigma}} \, dv, \\ \underline{\mathbf{R}}_\phi^I &= - \int_{\mathcal{B}_e} (\underline{\mathbf{B}}_\phi^I)^T \underline{\mathbf{B}} \, dv \\ \text{and} \quad \underline{\mathbf{R}}_\vartheta^I &= - \int_{\mathcal{B}_e} \left[N_\vartheta^I \left(\underline{\mathbf{L}} \underline{\dot{\boldsymbol{\vartheta}}} + \frac{\partial \mathcal{H}}{\partial \underline{\boldsymbol{\vartheta}}} \right) + (\underline{\mathbf{B}}_\vartheta^I)^T \underline{\boldsymbol{\Pi}} \right] dv. \end{aligned} \right\} \quad (3.14)$$

Note that the surface terms in (3.14) can be integrated by applying the boundary conditions in the finite-element software. Thereby, they are here ignored. The determination of the boundary conditions on the order parameter is not a trivial task. In the following, we will tacitly assume homogeneous Neumann type boundary conditions, i.e. $\vartheta_{\mu,j} n_j = \hat{\Theta}_\mu = 0$. Note that this Neumann

boundary automatically leads to the boundary $m_{i,j}n_j = A_{\mu i}(\vartheta_{\mu,j}n_j) = 0$ which is widely used in the conventional phase field model.

With respect to the time dependence of the residual, we use the implicit backward Euler time integration $\underline{\mathbf{d}} = (\underline{\mathbf{d}}_{n+1} - \underline{\mathbf{d}}_n)/\Delta t$. The quantities at the previous time-step t_n are denoted by $\underline{\mathbf{R}}_n^I$ and $\underline{\mathbf{d}}_n^I$. The equation for the current time-step t_{n+1}

$$\underline{\mathbf{R}}_{n+1}^I = \underline{\mathbf{R}}^I \left(\underline{\mathbf{d}}_n^I, \frac{\underline{\mathbf{d}}_{n+1}^I - \underline{\mathbf{d}}_n^I}{\Delta t} \right) \quad (3.15)$$

should be solved to obtain $\underline{\mathbf{d}}_{n+1}^I$. For solving these nonlinear equations, the Newton iteration scheme is performed at each time step. The corresponding iteration matrix for one element is

$$\underline{\mathbf{S}}^{IJ} = \underline{\mathbf{K}}^{IJ} + \frac{1}{\Delta t} \underline{\mathbf{D}}^{IJ}. \quad (3.16)$$

From the residuals, the stiffness matrix $\underline{\mathbf{K}}^{IJ}$ and the damping matrix $\underline{\mathbf{D}}^{IJ}$ can be calculated by derivation with respect to $\underline{\mathbf{u}}^I$, ϕ^I , ϑ^I and $\underline{\mathbf{u}}^I$, ϕ^I , ϑ^I , respectively. Specifically, the non-zero stiffness matrix can be derived as

$$\underline{\mathbf{K}}_{uu}^{IJ} = -\frac{\partial \underline{\mathbf{R}}_u^I}{\partial \underline{\mathbf{u}}^I} = \int_{\mathcal{B}_e} (\underline{\mathbf{B}}_u^I)^T \underline{\mathbf{C}} \underline{\mathbf{B}}_u^I \, dv, \quad \underline{\mathbf{K}}_{u\vartheta}^{IJ} = -\frac{\partial \underline{\mathbf{R}}_u^I}{\partial \underline{\vartheta}^I} = -\int_{\mathcal{B}_e} (\underline{\mathbf{B}}_u^I)^T \underline{\mathbf{C}} \frac{\partial \underline{\boldsymbol{\varepsilon}}^0}{\partial \underline{\vartheta}^I} N_{\vartheta}^I \, dv \quad (3.17)$$

$$\underline{\mathbf{K}}_{\phi\phi}^{IJ} = -\frac{\partial \underline{\mathbf{R}}_{\phi}^I}{\partial \phi^I} = -\int_{\mathcal{B}_e} (\underline{\mathbf{B}}_{\phi}^I)^T \underline{\mathbf{B}}_{\phi}^I \, dv, \quad \underline{\mathbf{K}}_{\phi\vartheta}^{IJ} = -\frac{\partial \underline{\mathbf{R}}_{\phi}^I}{\partial \vartheta^I} = \int_{\mathcal{B}_e} (\underline{\mathbf{B}}_{\phi}^I)^T \underline{\mathbf{M}} \underline{\mathbf{A}}^T N_{\vartheta}^I \, dv \quad (3.18)$$

$$\left. \begin{aligned} \underline{\mathbf{K}}_{\vartheta u}^{IJ} &= -\frac{\partial \underline{\mathbf{R}}_{\vartheta}^I}{\partial \underline{\mathbf{u}}^I} = -\int_{\mathcal{B}_e} N_{\vartheta}^I \left(\frac{\partial \underline{\boldsymbol{\varepsilon}}^0}{\partial \underline{\vartheta}^I} \right)^T \underline{\mathbf{C}} \underline{\mathbf{B}}_u^I \, dv, \\ \underline{\mathbf{K}}_{\vartheta\phi}^{IJ} &= -\frac{\partial \underline{\mathbf{R}}_{\vartheta}^I}{\partial \phi^I} = \int_{\mathcal{B}_e} N_{\vartheta}^I \underline{\mathbf{M}} \underline{\mathbf{A}} \underline{\mathbf{B}}_{\phi}^I \, dv \\ \underline{\mathbf{K}}_{\vartheta\vartheta}^{IJ} &= -\frac{\partial \underline{\mathbf{R}}_{\vartheta}^I}{\partial \vartheta^I} = \int_{\mathcal{B}_e} \left[N_{\vartheta}^I \left(\frac{\partial (\underline{\mathbf{L}} \dot{\vartheta})}{\partial \vartheta^I} + \frac{\partial^2 \mathcal{H}}{\partial \vartheta \partial \vartheta^I} \right) + (\underline{\mathbf{B}}_{\vartheta}^I)^T \frac{\partial \underline{\boldsymbol{\Pi}}}{\partial \vartheta^I} \right] \, dv \end{aligned} \right\} \quad (3.19)$$

and

and the damping matrix as

$$\underline{\mathbf{D}}_{\vartheta\vartheta}^{IJ} = -\frac{\partial \underline{\mathbf{R}}_{\vartheta}^I}{\partial \vartheta^I} = \int_{\mathcal{B}_e} N_{\vartheta}^I N_{\vartheta}^I \underline{\mathbf{L}} \, dv. \quad (3.20)$$

All other components of the damping matrix are zero. Note that the stiffness matrix components $\underline{\mathbf{K}}_{\vartheta\vartheta}^{IJ}$ and the damping matrix components $\underline{\mathbf{D}}_{\vartheta\vartheta}^{IJ}$ are unsymmetric. This unsymmetric characteristic is intrinsically attributed to the precession nature of magnetization dynamics. The elements are all integrated using a standard eight point Gauss integration scheme. It should be noted that this procedure is in the element level. For the whole simulated object, the above nonlinear equations and iteration matrix should be assembled into the global ones. The model is implemented as a user element in the software FEAP [32].

4. Simulation results

As a typical magnetostrictive material, $\text{Fe}_{81.3}\text{Ga}_{18.7}$ is simulated by using the constraint-free phase field model and its finite-element implementation. The material parameters are taken from the literature [14,33] and are listed in table 1. For the simulation of precession in §4a, a single element test was used. For the study of domains, a thin film nanostructure is simulated, in order to obtain the characteristic features and simultaneously save computation time. The sample mesh is $10 \times 20 \times 1$. Given that the exchange coefficient of $\text{Fe}_{81.3}\text{Ga}_{18.7}$ has a similar magnitude as Fe ($A_e \sim 10^{-11} \text{ J m}^{-1}$), the original size of the simulated sample is around $30 \times 60 \times 3 \text{ nm}^3$. The finite-element mesh size is chosen in a manner that a smooth variation of the magnetization over the whole sample should be resolved by the discretization. This means that the obtained

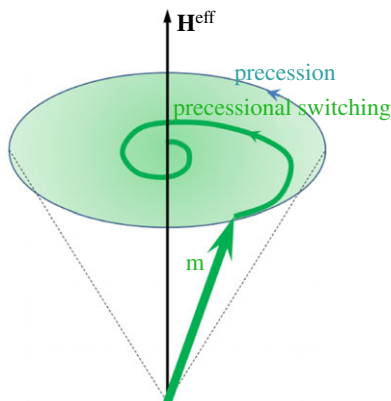


Figure 1. Illustration on precession and precessional switching of magnetization. (Online version in colour.)

Table 1. Material parameters of $\text{Fe}_{81.3}\text{Ga}_{18.7}$.

parameter	name	original value	dimensionless value
C_{11}	elastic constant	196 GPa	9.8×10^6
C_{12}	elastic constant	156 GPa	7.8×10^6
C_{44}	elastic constant	123 GPa	6.15×10^6
K_1	anisotropy coefficient	$2 \times 10^4 \text{ J m}^{-3}$	1.0
K_2	anisotropy coefficient	$-4.5 \times 10^4 \text{ J m}^{-3}$	-2.25
A_e	exchange coefficient	$\sim 10^{-11} \text{ J m}^{-1}$	1.0
M	saturation magnetization	$1.432 \times 10^6 \text{ A m}^{-1}$	11.348
λ_{100}	magnetostrictive constant	2.64×10^{-4}	2.64×10^{-4}
λ_{111}	magnetostrictive constant	0	0
μ_0	vacuum permeability	$4\pi \times 10^{-7} \text{ H m}^{-1}$	1.0

domain wall should span over at least two or more elements. Note that continuum models have been often used to study ferromagnetic domains in nanostructures [10,13,15,17,23]. As it will be shown by the comparison with experimental observations in §4b, the choice of the sample size $30 \times 60 \times 3 \text{ nm}^3$ and the application of our continuum phase field model are legitimate. For all the examples, the boundary condition $\vartheta_{\mu,j}n_j = 0$ is used. This leads to the boundary condition $m_{i,j}n_j = A_{\mu i}(\vartheta_{\mu,j}n_j) = 0$, which is commonly assumed in the literature (e.g. [8,13,15,23]).

(a) Precession and precessional switching

Precession and precessional switching are basic physical phenomena in magnetization dynamics [34]. If there is no damping (i.e. no dissipation), the instantaneous change in \mathbf{m} should be perpendicular to both the external field and the direction of \mathbf{m} . In other words, the effective magnetic field (\mathbf{H}^{eff}) is unable to rotate \mathbf{m} towards its direction, and \mathbf{m} rotates itself around \mathbf{H}^{eff} with a constant angle, as it is shown by the thin line in figure 1. This phenomenon is the so-called precession. If there is damping, \mathbf{m} can be gradually rotated towards the direction of \mathbf{H}^{eff} and forms a spiral-type path, as it is illustrated by the thick green line in figure 1. This is called as precessional switching. The spiral-type path can vary with the damping coefficient.

As a benchmark example, the constraint-free phase field model is used to reproduce precession and precessional switching. To simulate this behaviour of a monodomain, a single finite element

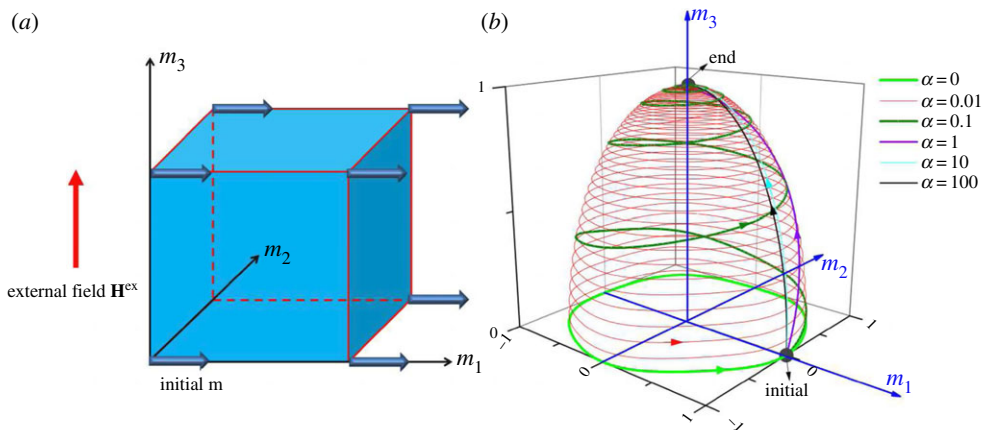


Figure 2. (a) Initial magnetization configuration and the external field for one-element simulation. (b) Simulated precession behaviour and damping-dependent precessional switching. (Online version in colour.)

should be sufficient. As shown in figure 2a, the initial magnetization is in the x_1 -direction, while the external magnetic field \mathbf{H}^{ex} is applied in the x_3 -direction. A field of $H^{\text{ex}} = 1.0$ is exerted by applying a magnetic potential difference to the two surfaces perpendicular to the x_3 -direction. The simulated rotation path of \mathbf{m} is presented in figure 2b, for different damping coefficient α . In the case of $\alpha = 0$, namely no damping, \mathbf{m} rotates around \mathbf{H}^{ex} , and the rotation trajectory is an in-plane circle perpendicular to the x_3 -direction. It indicates that the precession behaviour is reproduced. In the cases of $\alpha > 0$, damping dependent precessional switching is observed. When α is very small, for example $\alpha = 0.01$, a large number of precessional loops occur till \mathbf{m} is oriented along the direction of \mathbf{H}^{ex} . When α is moderately small, for example $\alpha = 0.1$, much less number of loops is required. When α is relatively large, for example $\alpha = 1.0$, 10 and 100, less than one precessional loop is needed, and \mathbf{m} directly rotates towards \mathbf{H}^{ex} . The rotation trajectory lies nearly in one plane. These simulation results of precession and precessional switching indicate that magnetization switching dynamics has been soundly considered in the constraint-free phase field model.

(b) Domain and magnetic vortex

The proposed model is further applied to simulate the domain structure in a thin nanostructure which is free from mechanical and magnetic load, as shown in figure 3. The mechanical boundary condition for all the sample surfaces is traction-free, i.e. $\boldsymbol{\sigma}^* \cdot \mathbf{n} = \mathbf{0}$. The magnetic boundary is set as $\mathbf{B}^* \cdot \mathbf{n} = 0$ and the damping coefficient $\alpha = 1.0$. We start with an initial random distribution shown in figure 3a and then perform finite-element simulation till the equilibrium state is obtained. As it is shown in figure 3b, two vortices are formed in the equilibrium configuration. The vortex, a curling configuration with a core, consists of four domains separated by 90° domain walls. Most interestingly, in the vortex core, the magnetization rotates gradually out of the plane. This phenomenon makes our simulation results essentially different from the vortices given in the literature [23,25]. It should be noted that this simulated vortex with curling and out-of-plane structure is attributed to the constraint on magnetization magnitude. In the vortex core, due to the strictly satisfied constraint $\|\mathbf{m}\| = 1$, the magnetization cannot vanish (must be unit) and turns perpendicular to the surface, thus avoiding the singularity. This is different in the case of ferroelectrics, where the polarization has no constraint in its magnitude and it can remain in-plane by reducing its magnitude to very small values [20]. In fact, with $\|\mathbf{m}\| = 1$, magnetization in the vortex remaining in plane would significantly increase both the exchange and the magnetocrystalline anisotropy energy. As the out-of-plane m_2 -direction is also

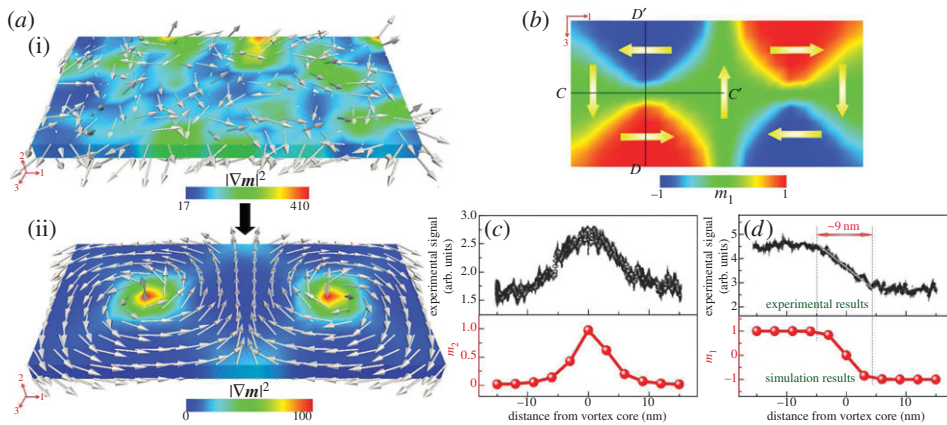


Figure 3. (a) Initial (i) and equilibrium (ii) magnetization configuration in a free-standing sample. The magnetic boundary condition is $\mathbf{B}^* \cdot \mathbf{n} = 0$. (b) The contour plot of m_1 in the equilibrium state. (c) Comparison of the measured and the calculated out-of-plane component m_2 along the line CC' shown in (b). (d) Comparison of the measured and the calculated in-plane component m_1 along the line DD' . Both CC' and DD' go through the vortex core. The experimental signal is taken from ref. [35]. (Online version in colour.)

the easy axis, the curling and twisting-out structure largely decreases the exchange energy and the magnetocrystalline anisotropy energy and thus is energetically favourable. Note that we have also considered a larger sample with size of $30 \times 60 \times 9 \text{ nm}^3$ and found that the domain configuration has no significant difference.

This simulated vortex phenomenon is consistent with the experimental observations in nanoscale ferromagnetic thin films [35]. Figure 3c shows the comparison of the out-of-plane component m_2 along the line CC' shown in figure 3b. A good agreement can be seen between the measured and the calculated results. Furthermore, the in-plane component m_1 along the line DD' is also considered. As demonstrated in figure 3d, the experimental and the simulated results agree well with each other. In particular, the measured and the calculated profiles show a vortex width of around 9 nm, also in accordance with the experimental measurements [35].

(c) Evolution of the vortex configuration under magnetic loading

Simulation is performed to show domain poling under a magnetic field. As an example, the equilibrium state in figure 3a is taken as the initial configuration. The external magnetic field $H^{\text{ex}*} = 1.0$ is applied antiparallel to the x_3 -direction. The damping coefficient is set as 1.0. As it is shown in figure 4a, the two vortices move oppositely towards the sample corners. The motion trajectory of the vortices is approximately along the in-plane diagonal of the sample. After these vortices vanish at the corners, a uniform configuration is formed gradually. The whole evolution procedure can also be demonstrated through the contour plot of the component m_3 , as it is shown in figure 4b. During this process, the regions with magnetization parallel to the external magnetic field expand gradually, whereas the regions with magnetization antiparallel and perpendicular to the external magnetic field shrink. It should be noted that the moving direction of the vortex has the component perpendicular to the external magnetic field, which is consistent with the experimental observation [36].

(d) 180° domain wall evolution under magnetic loading

As another example, the evolution of a 180° domain wall under an external magnetic field is investigated. As reported in §4a, the magnetization switching is dependent on the damping coefficient α . It has been even evidenced that the damping-induced dissipation is the driving

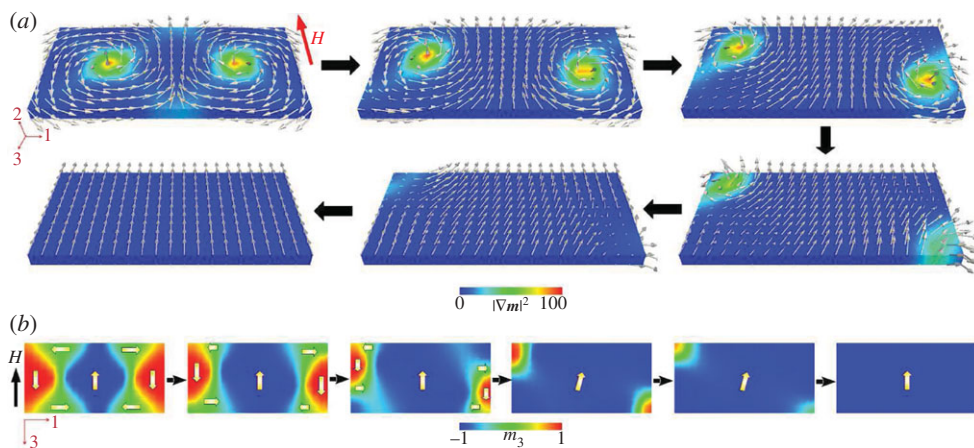


Figure 4. (a) Domain evolution under an external magnetic field, from the initial vortices configuration to a monodomain configuration. (b) Contour plot of m_3 during the evolution process shown in (a). (Online version in colour.)

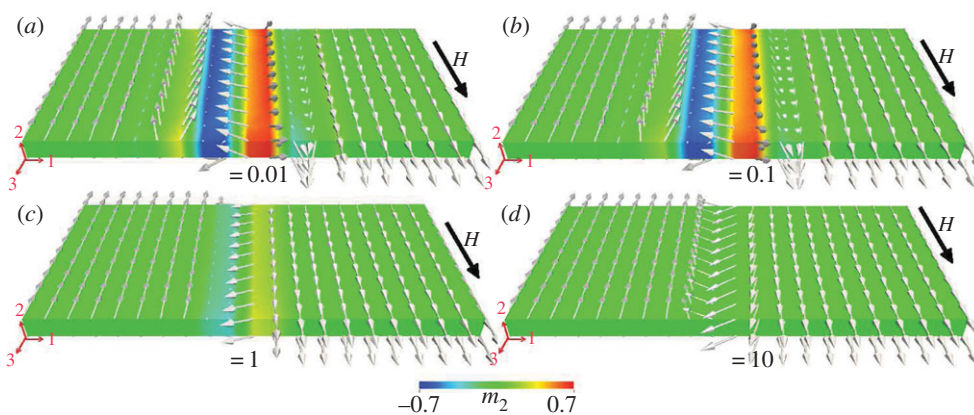


Figure 5. Magnetization distribution and contour plot of the out-of-plane component m_2 at $t^* = 5 \times 10^{-5}$ for the case (a) $\alpha = 0.01$, (b) $\alpha = 0.1$, (c) $\alpha = 1.0$ and (d) $\alpha = 10.0$. (Online version in colour.)

force for magnetic domain wall motion [37]. If there is no damping, the motion of the domain wall will not occur. In this subsection, we show that the magnetization switching of the 180° domain configuration can also be damping dependent. Figure 5 shows the distribution of out-of-plane magnetization m_2 . When the damping is small, e.g. $\alpha = 0.01, 0.1$, the magnetization near the 180° wall rotates out of the plane, as it is shown in figure 5a,b. Whereas, it can be seen that when the damping is large, e.g. $\alpha = 1.0, 10.0$, the rotation of magnetization is almost constrained in the plane, as shown in figure 5c,d. Intrinsically, this behaviour is attributed to the magnetization dynamics. By a LLG-type evolution equation which has only a damping term and no precession term, Miehe *et al.* showed in-plane rotation of magnetization [25]. Wang *et al.* used the TDGL equation (not the LLG equation) that intrinsically cannot involve the damping effect and found that most magnetization vectors rotate in plane [15].

For a comparison, the evolution of the 180° wall in the cases of $\alpha = 0.1$ and 10.0 is specifically given in figure 6. In the case of $\alpha = 0.1$, the magnetization near the wall first rotates out of plane, destroying the sharp 180° wall configuration. Then the magnetization away from the wall also rotates out of plane, and two vortices initiate at the sample boundary. Subsequently, the vortex in the region whose magnetization is along the external magnetic field rapidly disappears. The

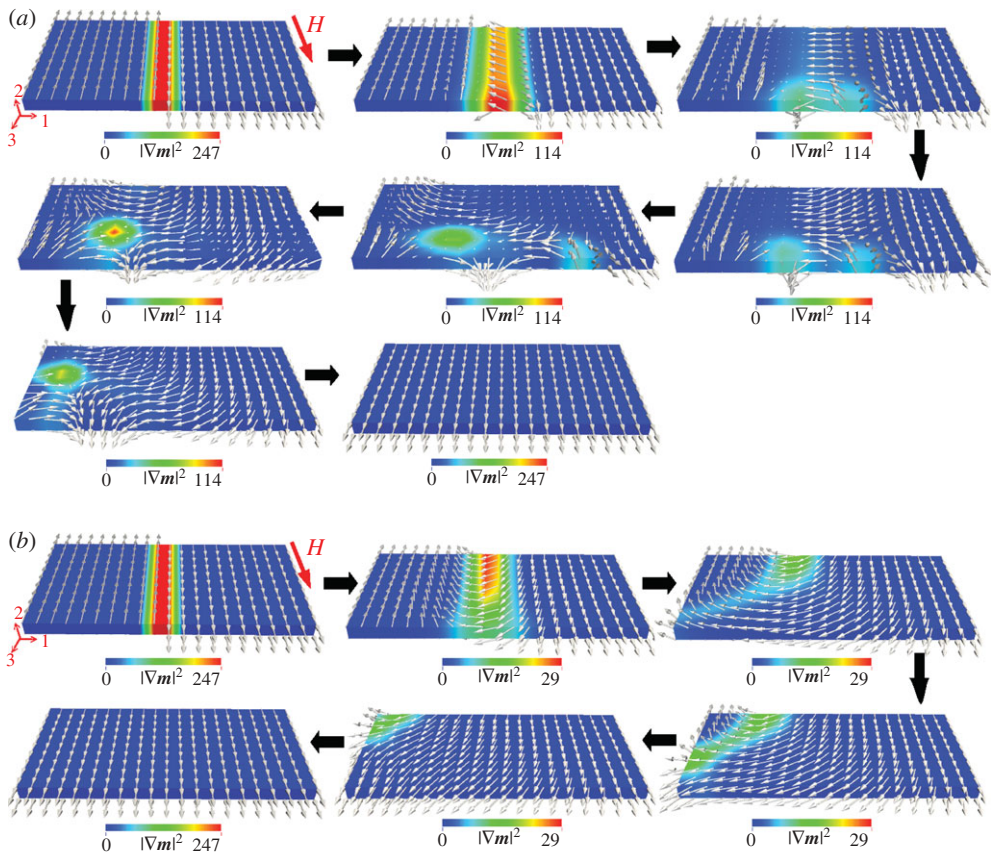


Figure 6. Poling of a 180° domain wall configuration for the case (a) $\alpha = 0.1$ and (b) $\alpha = 10.0$. (Online version in colour.)

other vortex forms and moves from one edge towards another edge under the drive of the external magnetic field. Finally, this vortex moves to the up-left corner and disappears there. The magnetization in the whole sample is then aligned along the the external magnetic field. Owing to the out-of-plane rotation of the magnetization and the formation of vortices, this evolution process is relatively complicated. During this process, the 180° wall was totally destroyed, and new magnetization configurations were constructed. By contrast, the evolution process in the case of large damping $\alpha = 10.0$ is much simpler, as one can see in figure 6*b*. The magnetization merely rotates in the plane and the 180° wall becomes diffusive. The magnetization in the left region, initially antiparallel to the external magnetic field, always rotates counterclockwise to be aligned along the external magnetic field. While the magnetization in the right region, initially parallel to the external magnetic field, rotates firstly clockwise and then counterclockwise. In summary, the poling process of a 180° domain configuration is strongly dependent on the damping coefficient α . It should be noted that a thicker sample with a mesh of $10 \times 20 \times 3$ is also calculated and it is found that the evolution process is almost the same.

(e) Domain evolution under mechanical loading

Besides the ferromagnetic switching, the constraint-free phase field model can also simulate the ferroelastic switching. It essentially makes this phase field model different from the micromagnetic models. In the simulation, the equilibrium vortex structure is taken as the initial configuration, as shown in figure 7*a*. The mechanical load is applied by assigning displacement. The surface at $x_3^* = 0$ is fixed, and the displacement at the opposite surface is set as $u_3^* = 0.07$ for

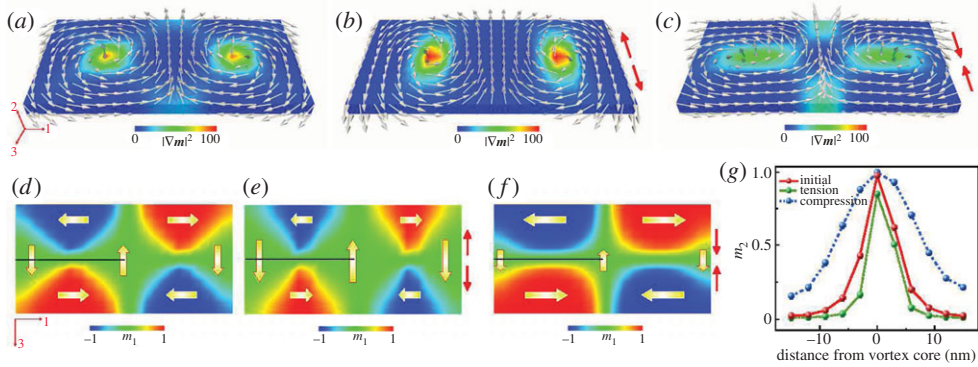


Figure 7. Mechanically driven domain evolution. (a) Initial magnetization configuration. Equilibrium states under (b) a tensile stress and (c) a compressive stress along the x_3 -direction. The domain configuration for the case of (d) initial state, (e) tensile stress and (f) compressive stress. (g) The change of m_2 along the horizontal lines in (d–f). (Online version in colour.)

the case of tension and $u_3^* = -0.07$ for the case of compression. The damping coefficient is 1.0. It is shown in figure 7b that the tensile loading causes the circular vortices to be elliptical, with the major axis along the tensile direction. As for the domains, the tensile loading increases the fraction of domains with the magnetization along the x_3 -direction at the expense of domains with the magnetization along the x_1 -direction, as shown in figure 7e. On the contrary, under the compressive loading, elliptical vortices with major axis perpendicular to the compressive direction appear (figure 7c). The fraction of domains with magnetization along the x_1 -direction remarkably increases (figure 7f). These results are expected. A tensile stress increases the regions of domains with magnetization parallel to the x_3 -direction, whereas a compressive stress decreases them. By examining the out-of-plane magnetization component m_2 along the line through the vortex, we can quantitatively obtain the changes in vortex, as shown in figure 7g. With respect to the x_1 -direction, the vortex becomes narrower under tensile loading while wider under compressive loading.

5. Conclusion

In conclusion, we have proposed a continuum constraint-free phase field model for ferromagnetic materials. Unlike conventional phase field models which take $\mathbf{m}(m_1, m_2, m_3)$ as the order parameters, the constraint-free phase field model uses the polar and azimuthal angles $(\vartheta_1, \vartheta_2)$. As a result, the constraint on magnetization is satisfied automatically within the model itself, and no additional numerical strategy for the phase field evolution is needed. The model was developed in a thermodynamic framework. Based on a configurational force system for $(\vartheta_1, \vartheta_2)$ and the second law of thermodynamics, a set of thermodynamically consistent constitutive relations and a generalized evolution equation for the order parameters $(\vartheta_1, \vartheta_2)$ have been obtained. As it has been shown in §3 that the model leads to a straightforward three-dimensional finite-element implementation, which requires no numerical treatment on the constraint and has one less degree of freedom.

The presented numerical simulations evidence that the model can readily predict the fundamental phenomenon in ferromagnetic domain switching. Reproduction of precession and damping-dependent precessional switching behaviour shows that the model gives a physically sound switching dynamics. Results on domain structure in a free-standing thin film demonstrates that the magnetization rotates around the vortex core and gradually twists out of the plane. The magnetization distribution around the vortex and the vortex width agree well with experimental observations. Under an external magnetic field, the vortex motion has a component perpendicular to the magnetic field, which is also consistent with experimental observations.

Ferromagnetic switching of 180° domain configuration is damping-dependent. Magnetization rotates out of plane and a vortex tends to form in the case of small damping, whereas magnetization is constrained into in-plane rotation in the case of large damping. Examples have been given for ferroelastic switching, i.e. domain evolution under mechanical loading. It is found that a tensile loading increases the fraction of domains along the tensile direction, whereas a compressive loading increases the fraction of domains perpendicular to the compressive direction. Accordingly, the vortex will change from a circular shape to an elliptical shape.

It can be concluded from the simulations that a three-dimensional model and implementation is indispensable, and the switching mechanism of magnetization is spatial. Furthermore, the ferromagnetic domain evolution is strongly damping dependent. These two features have not been well considered in the conventional phase field models in the literature. In our future work, the mesoscopic response of ferromagnetic materials will be investigated by using this model.

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Appendix A

With $\dot{m}_i = A_{\mu i} \dot{\vartheta}_\mu$ and $\mathbf{m} = [\sin \vartheta_1 \cos \vartheta_2 \quad \sin \vartheta_1 \sin \vartheta_2 \quad \cos \vartheta_1]^T$, the LLG equation (2.30) can be written as

$$\begin{aligned} \begin{bmatrix} \dot{\vartheta}_1 \cos \vartheta_1 \cos \vartheta_2 - \dot{\vartheta}_2 \sin \vartheta_1 \sin \vartheta_2 \\ \dot{\vartheta}_1 \cos \vartheta_1 \sin \vartheta_2 + \dot{\vartheta}_2 \sin \vartheta_1 \cos \vartheta_2 \\ -\dot{\vartheta}_1 \sin \vartheta_1 \end{bmatrix} &= -\gamma_0 \mu_0 \begin{bmatrix} \sin \vartheta_1 \cos \vartheta_2 \\ \sin \vartheta_1 \sin \vartheta_2 \\ \cos \vartheta_1 \end{bmatrix} \times \begin{bmatrix} H_1^{\text{eff}} \\ H_2^{\text{eff}} \\ H_3^{\text{eff}} \end{bmatrix} \\ &+ \alpha \begin{bmatrix} \sin \vartheta_1 \cos \vartheta_2 \\ \sin \vartheta_1 \sin \vartheta_2 \\ \cos \vartheta_1 \end{bmatrix} \times \begin{bmatrix} \dot{\vartheta}_1 \cos \vartheta_1 \cos \vartheta_2 - \dot{\vartheta}_2 \sin \vartheta_1 \sin \vartheta_2 \\ \dot{\vartheta}_1 \cos \vartheta_1 \sin \vartheta_2 + \dot{\vartheta}_2 \sin \vartheta_1 \cos \vartheta_2 \\ -\dot{\vartheta}_1 \sin \vartheta_1 \end{bmatrix} \\ &= -\gamma_0 \mu_0 \begin{bmatrix} \sin \vartheta_1 \sin \vartheta_2 H_3^{\text{eff}} - \cos \vartheta_1 H_2^{\text{eff}} \\ \cos \vartheta_1 H_1^{\text{eff}} - \sin \vartheta_1 \cos \vartheta_2 H_3^{\text{eff}} \\ \sin \vartheta_1 \cos \vartheta_2 H_2^{\text{eff}} - \sin \vartheta_1 \sin \vartheta_2 H_1^{\text{eff}} \end{bmatrix} \\ &+ \alpha \begin{bmatrix} -\dot{\vartheta}_1 \sin \vartheta_2 - \dot{\vartheta}_2 \sin \vartheta_1 \cos \vartheta_1 \cos \vartheta_2 \\ \dot{\vartheta}_1 \cos \vartheta_2 - \dot{\vartheta}_2 \sin \vartheta_1 \cos \vartheta_1 \sin \vartheta_2 \\ \dot{\vartheta}_2 \sin^2 \vartheta_1 \end{bmatrix}. \end{aligned} \quad (\text{A } 1)$$

Left multiplying (A 1) with the matrix

$$\begin{bmatrix} -\cos \vartheta_1 \cos \vartheta_2 \sin \vartheta_1 & -\cos \vartheta_1 \sin \vartheta_2 \sin \vartheta_1 & \sin^2 \vartheta_1 \\ -\sin \vartheta_2 & \cos \vartheta_2 & 0 \end{bmatrix} \quad (\text{A } 2)$$

can give

$$\begin{aligned} \begin{bmatrix} \dot{\vartheta}_2 \sin \vartheta_1 \\ -\dot{\vartheta}_1 \sin \vartheta_1 \end{bmatrix} &= -\gamma_0 \mu_0 \begin{bmatrix} \cos \vartheta_1 \cos \vartheta_2 H_1^{\text{eff}} + \cos \vartheta_1 \sin \vartheta_2 H_2^{\text{eff}} - \sin \vartheta_1 H_3^{\text{eff}} \\ -\sin \vartheta_1 \sin \vartheta_2 H_1^{\text{eff}} + \sin \vartheta_1 \cos \vartheta_2 H_2^{\text{eff}} \end{bmatrix} + \alpha \begin{bmatrix} \dot{\vartheta}_1 \\ \sin^2 \vartheta_1 \dot{\vartheta}_2 \end{bmatrix} \\ &\Rightarrow \begin{bmatrix} \dot{\vartheta}_2 \sin \vartheta_1 - \alpha \dot{\vartheta}_1 \\ -\dot{\vartheta}_1 \sin \vartheta_1 - \alpha \sin^2 \vartheta_1 \dot{\vartheta}_2 \end{bmatrix} \\ &= -\gamma_0 \mu_0 \begin{bmatrix} \cos \vartheta_1 \cos \vartheta_2 H_1^{\text{eff}} + \cos \vartheta_1 \sin \vartheta_2 H_2^{\text{eff}} - \sin \vartheta_1 H_3^{\text{eff}} \\ -\sin \vartheta_1 \sin \vartheta_2 H_1^{\text{eff}} + \sin \vartheta_1 \cos \vartheta_2 H_2^{\text{eff}} \end{bmatrix} \end{aligned}$$

$$\begin{aligned}
&\Rightarrow \frac{1}{\gamma_0} \begin{bmatrix} \alpha & -\sin \vartheta_1 \\ \sin \vartheta_1 & \alpha \sin^2 \vartheta_1 \end{bmatrix} \begin{bmatrix} \dot{\vartheta}_1 \\ \dot{\vartheta}_2 \end{bmatrix} \\
&= \mu_0 \begin{bmatrix} \cos \vartheta_1 \cos \vartheta_2 & \cos \vartheta_1 \sin \vartheta_2 & -\sin \vartheta_1 \\ -\sin \vartheta_1 \sin \vartheta_2 & \sin \vartheta_1 \cos \vartheta_2 & 0 \end{bmatrix} \begin{bmatrix} H_1^{\text{eff}} \\ H_2^{\text{eff}} \\ H_3^{\text{eff}} \end{bmatrix} \\
&\Rightarrow \frac{1}{\gamma_0} L_{\mu\gamma} \dot{\vartheta}_\gamma = \mu_0 A_{\mu j} H_j^{\text{eff}}. \tag{A 3}
\end{aligned}$$

Incorporating the external magnetic field g_j^{ex}/μ_0 , the effective field can be derived as

$$H_j^{\text{eff}} = -\frac{1}{\mu_0 M} \frac{\delta \mathcal{H}}{\delta m_j} + \frac{g_j^{\text{ex}}}{\mu_0} = -\frac{1}{\mu_0 M} \left(\frac{\partial \mathcal{H}}{\partial m_j} - 2A_e m_{j,ii} \right) + \frac{g_j^{\text{ex}}}{\mu_0}. \tag{A 4}$$

Thus,

$$\begin{aligned}
\mu_0 A_{\mu j} H_j^{\text{eff}} &= -\frac{1}{M} A_{\mu j} \frac{\partial \mathcal{H}}{\partial m_j} + \frac{2A_e}{M} A_{\mu j} m_{j,ii} + A_{\mu j} g_j^{\text{ex}} \\
&= -\frac{1}{M} \frac{\partial \mathcal{H}}{\partial \vartheta_\mu} + \frac{2A_e}{M} (P_{\mu\gamma} \vartheta_{\gamma,ii} + P_{\mu\gamma,i} \vartheta_{\gamma,i}) + \zeta_\mu^{\text{ex}} \\
&= -\pi_\mu + \Pi_{\mu j,j} + \zeta_\mu^{\text{ex}}. \tag{A 5}
\end{aligned}$$

Combining (A 3) and (A 5) yields

$$\Pi_{\mu j,j} - \pi_\mu + \zeta_\mu^{\text{ex}} = \frac{1}{\gamma_0} L_{\mu\gamma} \dot{\vartheta}_\gamma \tag{A 6}$$

which is the same as the evolution equation in (2.28).

References

1. Goldman A. 1999 *Handbook of modern ferromagnetic materials*. New York, NY: Springer.
2. Comstock RL. 2002 Review modern magnetic materials in data storage. *J. Mater. Sci. Mater. Electron.* **13**, 509–523. (doi:10.1023/A:1019642215245)
3. Calkins FT, Flatau AB, Dapino MJ. 2007 Overview of magnetostrictive sensor technology. *J. Intell. Mater. Syst. Struct.* **18**, 1057–1066. (doi:10.1177/1045389X06072358)
4. Kittel C. 1949 Physical theory of ferromagnetic domains. *Rev. Mod. Phys.* **21**, 541–583. (doi:10.1103/RevModPhys.21.541)
5. Hubert A, Schäfer R. 1998 *Magnetic domains*. Berlin, Germany: Springer.
6. Wang J, Steinmann P. 2012 A variational approach towards the modeling of magnetic field-induced strains in magnetic shape memory alloys. *J. Mech. Phys. Solids* **60**, 1179–1200. (doi:10.1016/j.jmps.2012.02.003)
7. Landau LD, Lifshitz E. 1935 On the theory of the dispersion of magnetic permeability in ferromagnetic bodies. *Phys. Z. Sowjetunion* **8**, 153–169. See <http://ujp.bitp.kiev.ua/files/journals/53/si/53SI06p.pdf>.
8. Brown WF. 1963 *Micromagnetics*. New York, NY: Interscience Publishers.
9. Gilbert TL. 2004 A phenomenological theory of damping in ferromagnetic materials. *IEEE Trans. Magn.* **40**, 3443–3449. (doi:10.1109/TMAG.2004.836740)
10. Williams W, Dunlop DJ. 1989 Three-dimensional micromagnetic modelling of ferromagnetic domain structure. *Nature* **337**, 634–637. (doi:10.1038/337634a0)
11. Fidler J, Schrefl T. 2000 Micromagnetic modelling: the current state of the art. *J. Phys. D Appl. Phys.* **33**, R135–R156. (doi:10.1088/0022-3727/33/15/201)
12. Fidler J, Schrefl T, Scholz W, Suess D, Dittrich R, Kirschner M. 2004 Micromagnetic modelling and magnetization processes. *J. Magn. Magn. Mater.* **272–276**, 641–646. (doi:10.1016/j.jmmm.2003.12.1007)
13. Shu YC, Lin MP, Wu KC. 2004 Micromagnetic modeling of magnetostrictive materials under intrinsic stress. *Mech. Mater.* **36**, 975–997. (doi:10.1016/j.mechmat.2003.04.004)

14. Zhang JX, Chen LQ. 2005 Phase-field microelasticity theory and micromagnetic simulations of domain structures in giant magnetostrictive materials. *Acta Mater.* **53**, 2845–2855. (doi:10.1016/j.actamat.2005.03.002)
15. Wang J, Zhang J. 2013 A real-space phase field model for the domain evolution of ferromagnetic materials. *Int. J. Solids Struct.* **50**, 3597–3609. (doi:10.1016/j.ijsolstr.2013.07.001)
16. Koyama T. 2008 Phase-field modeling of microstructure evolutions in magnetic materials. *Sci. Technol. Adv. Mater.* **9**, 013006. (doi:10.1088/1468-6996/9/1/013006)
17. Hu J-M, Sheng G, Zhang JX, Nan CW, Chen LQ. 2011 Phase-field simulation of strain-induced domain switching in magnetic thin films. *Appl. Phys. Lett.* **98**, 112505. (doi:10.1063/1.3567542)
18. Landis CM. 2008 A continuum thermodynamics formulation for micro-magneto-mechanics with applications to ferromagnetic shape memory alloys. *J. Mech. Phys. Solids* **56**, 3059–3076. (doi:10.1016/j.jmps.2008.05.004)
19. Li LJ, Lei CH, Shu YC, Li JY. 2011 Phase-field simulation of magnetoelastic couplings in ferromagnetic shape memory alloys. *Acta Mater.* **59**, 2648–2655. (doi:10.1016/j.actamat.2011.01.001)
20. Xu BX, Schrade D, Müller R, Gross D. 2009 Micromechanical analysis of ferroelectric structures by a phase field method. *Comp. Mater. Sci.* **45**, 832–836. (doi:10.1016/j.commatsci.2008.07.010)
21. Schrade D, Müller R, Gross D, Keip M-A, Thai H, Schröder J. 2014 An invariant formulation for phase field models in ferroelectrics. *Int. J. Solids Struct.* **51**, 2144–2156. (doi:10.1016/j.ijsolstr.2014.02.021)
22. Yang B, Fredkin DR. 1998 Dynamical micromagnetics by the finite element method. *IEEE Trans. Magn.* **34**, 3842–3852. (doi:10.1109/20.728293)
23. Szabolics H, Buda-Prejbeanu LD, Toussaint JC, Fruchart O. 2008 A constrained finite element formulation for the Landau–Lifshitz–Gilbert equations. *Comp. Mater. Sci.* **44**, 253–258. (doi:10.1016/j.commatsci.2008.03.019)
24. Alouges F, Jaisson P. 2006 Convergence of a finite element discretization for the Landau–Lifshitz equations in micromagnetism. *Math. Models Methods Appl. Sci.* **16**, 299–316. (doi:10.1142/S0218202506001169)
25. Miehe C, Ethiraj G. 2012 A geometrically consistent incremental variational formulation for phase field models in micromagnetics. *Comput. Methods Appl. Mech. Eng.* **245–246**, 331–347. (doi:10.1016/j.cma.2012.03.021)
26. Ethiraj G, Miehe C. 2012 A projection method for phase field models in micromagnetics. *Proc. Appl. Math. Mech.* **12**, 399–400. (doi:10.1002/pamm.2012101881)
27. Krishnaprasad PS, Tan X. 2001 Cayley transforms in micromagnetics. *Physica B* **306**, 195–199. (doi:10.1016/S0921-4526(01)01003-1)
28. Fried E, Gurtin ME. 1994 Dynamic solid-solid transitions with phase characterized by an order parameter. *Phys. D* **72**, 287–308. (doi:10.1016/0167-2789(94)90234-8)
29. Steinmann P, Maugin GA. 2010 *Mechanics of material forces*. New York, NY: Springer.
30. Gross D, Kolling R, Müller R, Schmidt I. 2003 Configurational forces and their application in solid mechanics. *Eur. J. Mech. A/Solids* **22**, 669–692. (doi:10.1016/S0997-7538(03)00076-7)
31. Zienkiewicz OC, Taylor RL. 2000 *The finite element method volume 1: the basis*. Oxford, UK: Butterworth-heinemann.
32. *FEAP-A finite element analysis program*. See <http://www.ce.berkeley.edu/projects/feap/>.
33. Clark AE, Hathaway KB, Wun-Fogle M, Restorf JB, Lograsso TA, Keppens VM, Petculescu G, Taylor RA. 2003 Extraordinary magnetoelasticity and lattice softening in bcc Fe–Ga alloys. *J. Appl. Phys.* **93**, 8621–8623. (doi:10.1063/1.1540130)
34. Skomski R. 2008 *Simple models of magnetism*. Oxford, UK: Oxford University Press.
35. Wachowiak A, Wiebe J, Bode M, Pietzsch O, Morgenstern M, Wiesendanger R. 2002 Direct observation of internal spin structure of magnetic vortex cores. *Science* **298**, 577–580. (doi:10.1126/science.1075302)
36. Schneider M, Hoffmann H, Zweck J. 2000 Lorentz microscopy of circular ferromagnetic permalloy nanodisks. *Appl. Phys. Lett.* **77**, 2909–2911. (doi:10.1063/1.1320465)
37. Wang XR, Yan P, Lu J, He C. 2009 Magnetic field driven domain-wall propagation in magnetic nanowires. *Ann. Phys.* **324**, 1815–1820. (doi:10.1016/j.aop.2009.05.004)